



Electrochemical synthesis of TiO₂/Graphene oxide composite films for photocatalytic applications



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ABSTRACT

This paper presents a novel TiO₂-Graphene composite film one-step preparation method by electrochemical synthesis. The films were prepared on a Ti substrate by electrochemical deposition from a GO and Ti (IV) aqueous solution. For comparison, pure TiO₂ on a Ti substrate was obtained by the same electrochemical method. The as-prepared samples were characterized by X-ray diffraction, scanning electron microscopy, Raman spectroscopy, photoluminescence and ultraviolet–visible spectrophotometry. The photocatalytic properties of the films were investigated for the discoloration of Reactive Yellow 145 dye. The results showed that the TiO₂-Graphene oxide composite film presented photocatalytic activity almost twice as high as that observed for the TiO₂ film. In addition, the TiO₂-Graphene oxide composite film presented better stability and recyclability, retaining its catalytic efficiency ($\approx 70\%$) even after 15 cycles.

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1. Introduction

TiO₂ is one of the most important materials in the field of photocatalytic waste water purification [1–3]. In addition to its outstanding photocatalytic properties, TiO₂ presents good stability, is not toxic and has a low cost, making it one of the main components of the new materials developed for the photocatalytic treatment of waste water. The major limitation concerning its applicability is its narrow sun light range absorbance, which results in a very low efficiency [3–5]. When TiO₂ is irradiated by light with energy higher than its bandgap, an electron–hole pair is generated at the surface. The photon-induced electrons and holes have a strong oxidizing power, but their oxidation ability is limited by their fast recombination rates, leading to a low photocatalytic activity. In order to overcome these problems, several strategies have been employed. One of the most promising approaches to tackle

this limitation is the combination of TiO₂ with carbon nanomaterials, such as carbon nanotubes [6–8], fullerene C60 [8], graphene-based materials [8–11], and nanodiamonds [12].

Previous studies have demonstrated that combining TiO₂ with graphene oxide (GO) or reduced graphene oxide (rGO) can decrease the electron–hole recombination rate, thus lowering its gap energy and enhancing its photocatalytic performance [8,9,13,14]. However, many studies have shown that the photocatalytic activity of these composites is significantly affected by the material textural and morphological properties [9–12]. The preparation of GO/TiO₂ and rGO/TiO₂ composites usually involves complex experimental procedures, because of the poor interfacial contact of the oxide particles with the carbon surface and their tendency to agglomerate. Some studies claim that increasing the proportion of carbon in these composites may improve their photocatalytic activity by favoring the adsorption of the pollutant [15,16].

However, recent studies have shown that there is an optimum GO-TiO₂ ratio for maximum photocatalytic activity [8,9]. The carbon material concentration may affect the interaction between TiO₂ and rGO or GO particles, which is the main feature required to enhance charge separation and electron transfer. J. Yang et al.

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investigated TiO₂-Graphene microspheres prepared by ultrasonic spray pyrolysis and suggested that a higher graphene content may limit the contact of the TiO₂ surface with the pollutant and reduce the photocatalytic activity [16]. W. Jo et al. also investigated the photocatalytic activity of a GO/TiO₂ composite synthesized by a colloidal blending process [17] and also observed that a higher GO content in the composites decreased the degradation efficiency. More recently, the oxygen functionalities in GO have been found to be essential to mediate a uniform assembly of TiO₂ nanoparticles onto GO sheets with ensuing improved photocatalytic performance in comparison to rGO-based composites [11].

The present work reports a new single-step method to obtain TiO₂/GO composites by electrochemical synthesis. A Ti plate was used as a substrate for TiO₂/GO composite film preparation by electrochemical deposition from a GO-Ti(IV) aqueous solution. Ti(IV) was added to the solution using K₂TiF₆, which is soluble in water and therefore will produce the hydrolysis of Ti(IV) species. This strategy exploits the simultaneous deposition of TiO₂ and GO onto the substrate by cyclic voltammetry (CV). For comparison, pure TiO₂ on Ti substrate was obtained by the same electrochemical method, but without GO added to the electrolyte solution. The prepared TiO₂/GO composite and TiO₂ films were tested in the photocatalytic decolorization of Reactive Yellow 145, a dye used in the textile industry.

2. Experimental section

2.1. Graphene oxide preparation and purification

GO was prepared through oxidation of natural nanographite (99.5 wt% of purity), supplied by the *Brazilian Company Nacional de Grafite Ltda.* The oxidation procedure used H₂SO₄, KMnO₄ and H₂O₂, based on Hummers method [18]. The oxidation product was first washed with HCl (10%), then filtered and dried at 65 °C, and finally purified by subsequent washing with aqueous NaOH/water at pH 7, ethanol, and ether to remove oxidized carbonaceous residues. Finally, GO was filtered on a PTFE membrane with a pore diameter of 0.45 μm and dried for 24 h at 100 °C under vacuum [19].

2.2. Ti/TiO₂ and TiO₂/GO composite film preparation

A 2-mm thick pure Ti plate with a geometric area of 30 cm² was employed as a working electrode onto which the TiO₂ and graphene oxide (GO) film were electrochemically deposited. The Ti-plate surface was sequentially ground with a 360–2000-grit SiC paper, rinsed with acetone, ultrasonically cleaned in isopropyl alcohol for 20 min, and then preserved in isopropyl alcohol until further use. A platinum wire connected to the Ti-plate provided electrical contact to the working electrode. A three-electrode cell (200 mL) was used in all experiments. Platinum foil (30 cm²) was used as a counter electrode and an Ag/AgCl electrode was used as a reference. The CV experiments were performed using Autolab Electrochemical System (EcoChemie, Utrecht, The Netherlands) equipped with PGSTAT-12 and GPES software. In order to obtain the TiO₂/GO film, the electrochemical experiments were conducted using a mixture of 15 mg of GO and 100 mg of K₂TiF₆ in 150 mL of phosphate buffer at pH 9.0 as a support electrolyte. The sample obtained was labeled Ti/TiO₂/GO. The Ti/Ti/TiO₂ film was prepared using 100 mg of K₂TiF₆ in 150 mL phosphate buffer at pH 9.0 as a support electrolyte. The CV experiments were carried out over the potential range from –1.0–5.0 V at a scan rate of 50 mV s⁻¹. For comparison, a Ti/TiO₂ sample was prepared using only the phosphate buffer at pH 9.0 as a support electrolyte. Table 1 summarizes the preparation conditions of the prepared samples.

2.3. Characterization

Fourier-transform Raman spectroscopy was conducted using a Bruker RFS 100 instrument, a Nd³⁺/YAG laser operating in near-infrared at 1064 nm and a liquid N₂-cooled charge-coupled device detector. Good signal-to-noise ratios were obtained with 1000 scans accumulated over approximately 30 min and 4-cm⁻¹ spectral resolution. Surface morphology observations were obtained by scanning electron microscopy (SEM) with a Quanta 200 - FEG - FEI - 2006 unit. The samples were attached to a metal sample holder with adhesive carbon tape. X-Ray diffraction of the Ti-plate samples was performed by using a Shimadzu XRD-6000 diffractometer operating with λ = 0.15418 nm at 30 kV and 30 mA and a step size of 0.02°. Diffuse Reflectance Ultraviolet–Visible Spectroscopy (DRUV) was conducted on a Cary 5000-UV–Vis-NIR Spectrophotometer (Agilent Technologies) using an external diffuse reflectance accessory. Photoluminescence (PL) spectra were measured on a RF-5301PC Spectrofluorophotometer (Shimadzu). The samples were excited at the wavelength of 350 nm with a 150-W Xenon lamp under ambient conditions.

2.4. Photocatalytic tests

The photocatalytic performance of Ti/Ti/TiO₂ and Ti/TiO₂/GO composite films were evaluated in the discoloration of Reactive Yellow 145 using two 15-W ultraviolet–visible (UV–Vis) light lamps with spectral output from 250 to 1000 nm, and peak activity between 400 and 800 nm. A 150-mL reaction flask with a cooling jacket was placed 30 cm away from the light sources. The Ti substrate containing the composite film was placed into the reaction flask and one side (area of 30 cm²) was exposed to the light sources. All the experiments were performed using 70 mL of a synthetic aqueous solution of 10 mgL⁻¹ of Reactive Yellow 145 and textile dyeing bath samples containing around 8.5 mgL⁻¹ of the such dye. Before starting the photocatalytic reaction, the solution was stirred for 30 min in the dark to achieve adsorption equilibrium. After that, the light was switched on in order to start the photocatalytic reaction. The dye concentration was determined measuring the amount of light absorbed at 420 nm by means of a Cary-50 UV–Vis spectrophotometer. The photo-discoloration rate was evaluated by measuring the dye concentration as function of time in the first 10 min.

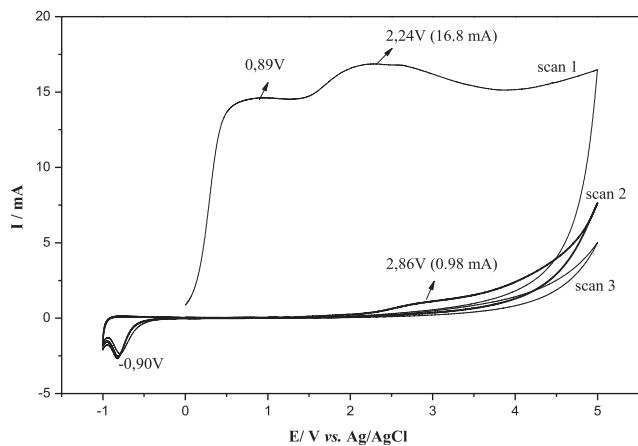
3. Results and discussion

Surface modification of the Ti substrate was detected through changes in the CV profile, displayed in Fig. 1(a–c). The similarity between the substrate CVs indicates the formation of TiO₂, which probably is the major electrochemical process that occurred in the electrode. Indeed, as soon as CV was performed between potentials –1.0 and +5.0 V, TiO₂ formation was demonstrated by the anodic current peaks centered at 0.89 V and 2.15 V [20,21]. Thus, the oxide layer began to form during anodic oxidation, inhibiting further Ti dissolution. Hence, after the first scan, the current decreased significantly due to electrode surface passivation by the oxide layer, which increased the Ti surface charge transfer resistance.

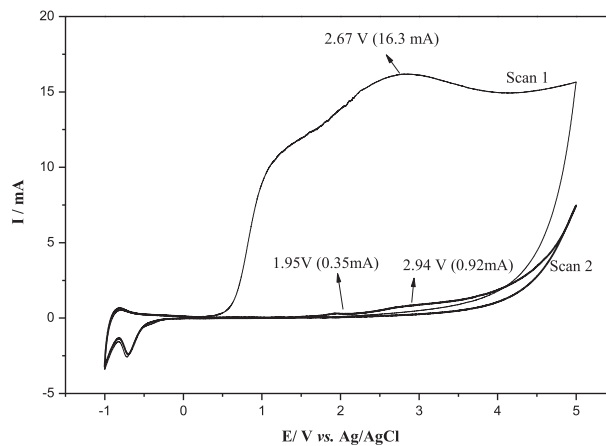
The effect of the addition of the Ti(IV) salt to the electrolyte can be observed in the CV of sample Ti/Ti/TiO₂ showed in Fig. 1(b). In this case, the anodic peak was observed mainly at a higher potential (2.67 V). The oxidation potential increase is reported in the literature as resulting from different oxide growth rates on different single grains on the titanium substrate, and also from the oxide phase transformation [22,23]. Indeed, although TiO₂ is more stable and frequently found in the oxide film, different Ti oxides can be

Table 1
Support electrolytes employed in the electrochemical preparation of each sample.

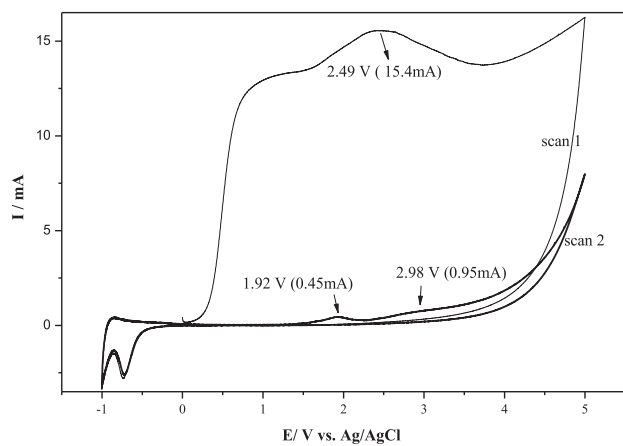
Sample	Support electrolyte
Ti/TiO ₂	K ₂ TiF ₆ solubilized in phosphate buffer at pH 9.0
Ti/TiO ₂ /GO	GO dispersion and K ₂ TiF ₆ solubilized in phosphate buffer at pH 9.0
Ti/TiO ₂	Phosphate buffer at pH 9.0



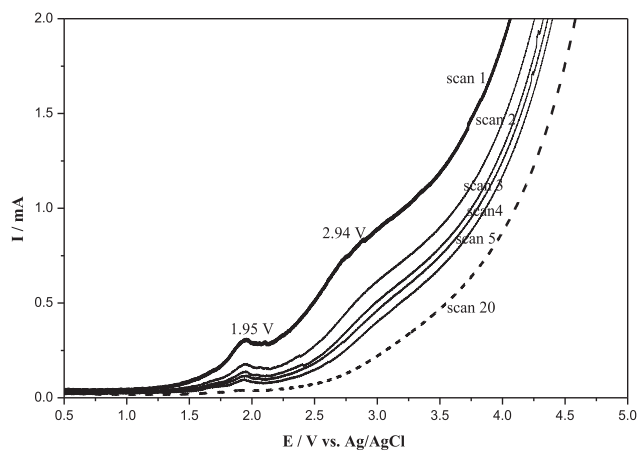
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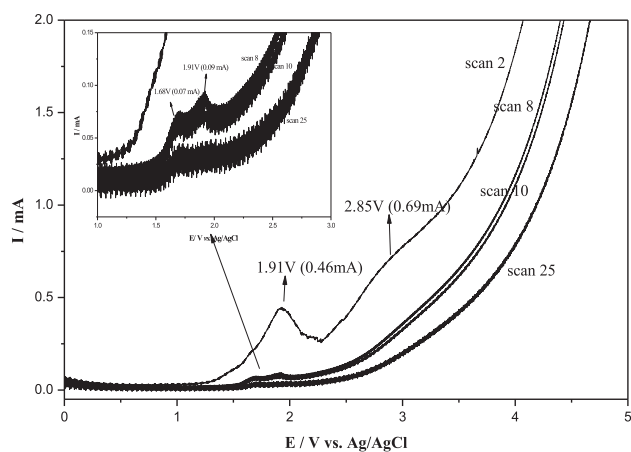
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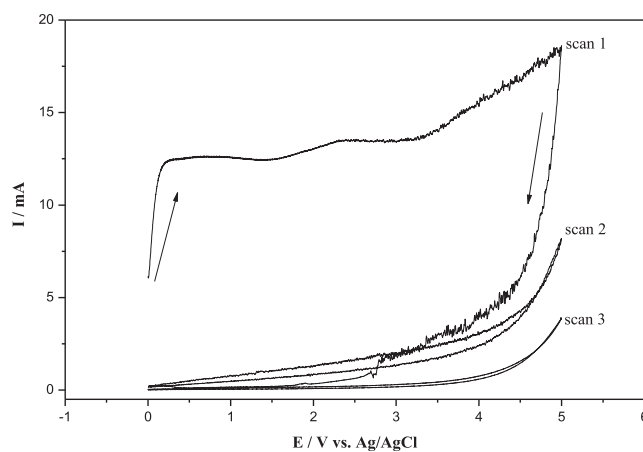
c)



d)



e)



f)

Fig. 1. Cyclic voltammograms recorded at 50 mV⁻¹ for TiO₂ growth in phosphate buffer solution at pH 9.0: a) Ti/TiO₂; b) Ti/Ti/TiO₂ and c) Ti/TiO₂/GO; d) Anodic CV peaks observed during the positive scan for Ti/Ti/TiO₂; e) Anodic CV peaks observed during the positive scan for Ti/TiO₂/GO and Inset: anodic curves for scans 8, 10 and 25 for Ti/TiO₂/GO and f) Cyclic voltammograms with cathodic sweep limited to 0V for Ti/Ti/TiO₂.

formed on the Ti surface during anodic oxidation, such as Ti_4O_7 , Ti_3O_5 and Ti_2O_3 . A similar behavior was observed in the CV for Ti/TiO₂/GO, shown in Fig. 1(c), which has a main anodic peak at 2.49 V. The Ti/TiO₂ and Ti/TiO₂/GO second scans showed two anodic peaks, one around 1.90 V and another poorly defined around 2.90 V. These peaks can be better observed in Fig. 1(d) and 1e. It can be noticed that the last anodic peak practically vanished in the other scans, while the first peak gradually decreased, but did not vanish with the number of scans. The presence of these anodic peaks even after several CV cycles suggests that several layers of oxides were formed in the process. Thus, the addition of K_2TiF_6 to the electrolyte produced hydrolyzed Ti(IV), contributing to the formation of an oxide layer denser than that observed for Ti/TiO₂.

The cathodic current peak observed at -0.73 V is attributed to the partial reduction of more superficial oxides and the peak at -1.00 V is attributed to the reduction of H^+ . This assignment is supported by the result shown in Fig. 1(f) for sample Ti/TiO₂, where by limiting the cathodic sweep to 0 V, the anodic peaks observed around 2.0 V and 3.0 V vanished.

In the case of Ti/TiO₂/GO, after the first scan, the anodic peak split into two other peaks, located at 1.68 V (0.07 mA) and 1.89 V

(0.09 mA), as shown in inset in Fig. 1(d). It seems that two different deposition processes took place on the electrode surface with the probable formation of two layers. The peak at a more positive potential (1.89 V) possibly originated from the TiO₂/GO film accumulated in a layer deeper than that at a less positive potential (1.68).

The surfaces of the as-prepared films on the Ti substrate were analyzed by X-ray diffraction; the results are shown in Fig. 2. The XRD patterns attributed to the Ti substrate occurred at about $2\theta = 35.2^\circ, 38.5^\circ, 40.4^\circ, 53.1^\circ, 63.1^\circ, 76.2^\circ$ and 77.5° corresponding to diffractions from the (100), (002), (101), (102), (110), (112) and (201) planes, respectively [24,25]. After CV cycling between potentials -1.0 V and $+5.0$ V, Ti/TiO₂ still presented peaks attributed to pure Ti. The absence of typical TiO₂ peaks may be related to low TiO₂ crystallization during the anodizing process. Additionally, the smaller thickness of the oxide film may lead to thin-film interference problems [25]. However, the XRD patterns for Ti/Ti/

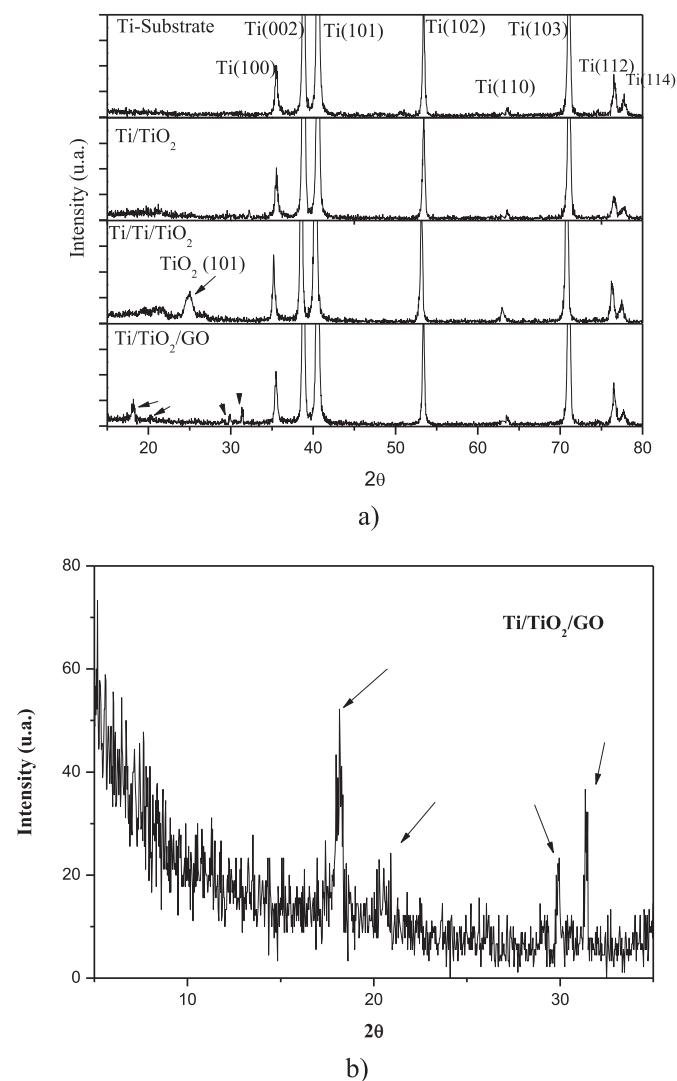


Fig. 2. a) XRD patterns of Ti, Ti/TiO₂, Ti/TiTiO₂ and Ti/TiO₂/GO; b) Magnified image of the Ti/TiO₂/GO XRD patterns.

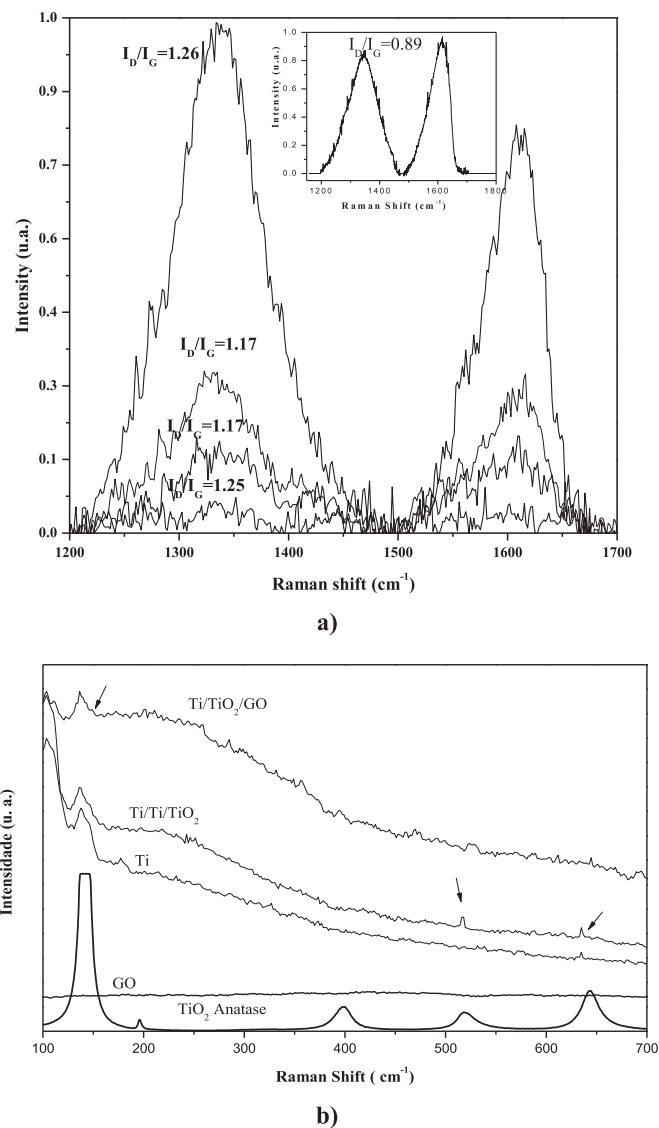


Fig. 3. Raman spectra of Ti/TiO₂/GO obtained a) in the range of 1200–1700 cm^{-1} , acquired at four different points of the surface and Inset: Raman spectra and I_D/I_G ratio value for the GO used as starting material to prepare the composite; b) in the range of 100–750 cm^{-1} in order to investigate the presence of TiO₂.

TiO₂ revealed one anatase peak at $2\theta = 25.3^\circ$, corresponding to diffractions from the (101) plane. These results are consistent with those observed for the CVs, which showed that the addition of Ti (IV) to the electrolyte can enhance the oxide layer formation. The Ti/TiO₂/GO XRD pattern (Fig. 2(b)) shows four poorly defined peaks at $2\theta = 18.2^\circ$, 20.6° , 29.6° and 31.2° , which could not be linked to any Ti oxides or GO typical crystal plane diffraction. This suggests that these peaks may be derived from new crystalline phases originated from the reaction of TiO₂ and GO during the anodizing process.

Raman spectroscopy is a very effective tool to investigate the detailed bonding structure of carbon-based nanostructures [26]. In this work, the surfaces of the as-prepared composite films were evaluated employing a conventional optical microscope coupled to

a Raman spectrometer. Consequently, it was possible to obtain information about the film composition at different points of the substrate surface. Fig. 3(a) shows the Raman spectra of Ti/TiO₂/GO obtained at four different points of the surface. The Raman spectra obtained in the range of $1200\text{--}1700\text{ cm}^{-1}$ display two typical vibration modes for carbon materials. These bands were identified as D and G bands, confirming the presence of GO in Ti/TiO₂/GO. The D-band, at $\sim 1339\text{ cm}^{-1}$, is usually observed for sp² carbons containing impurities or other symmetry-breaking defects, while the G-band, at $\sim 1609\text{ cm}^{-1}$, is related to vibrations in sp² carbon materials and is attributed to well-ordered graphite [27]. Thus, a quantitative measure of functionalization density in GO can be estimated by the ratio between the areas of the D (I_D) and G bands (I_G). The I_D/I_G ratio value observed for the GO used as starting material to prepare the

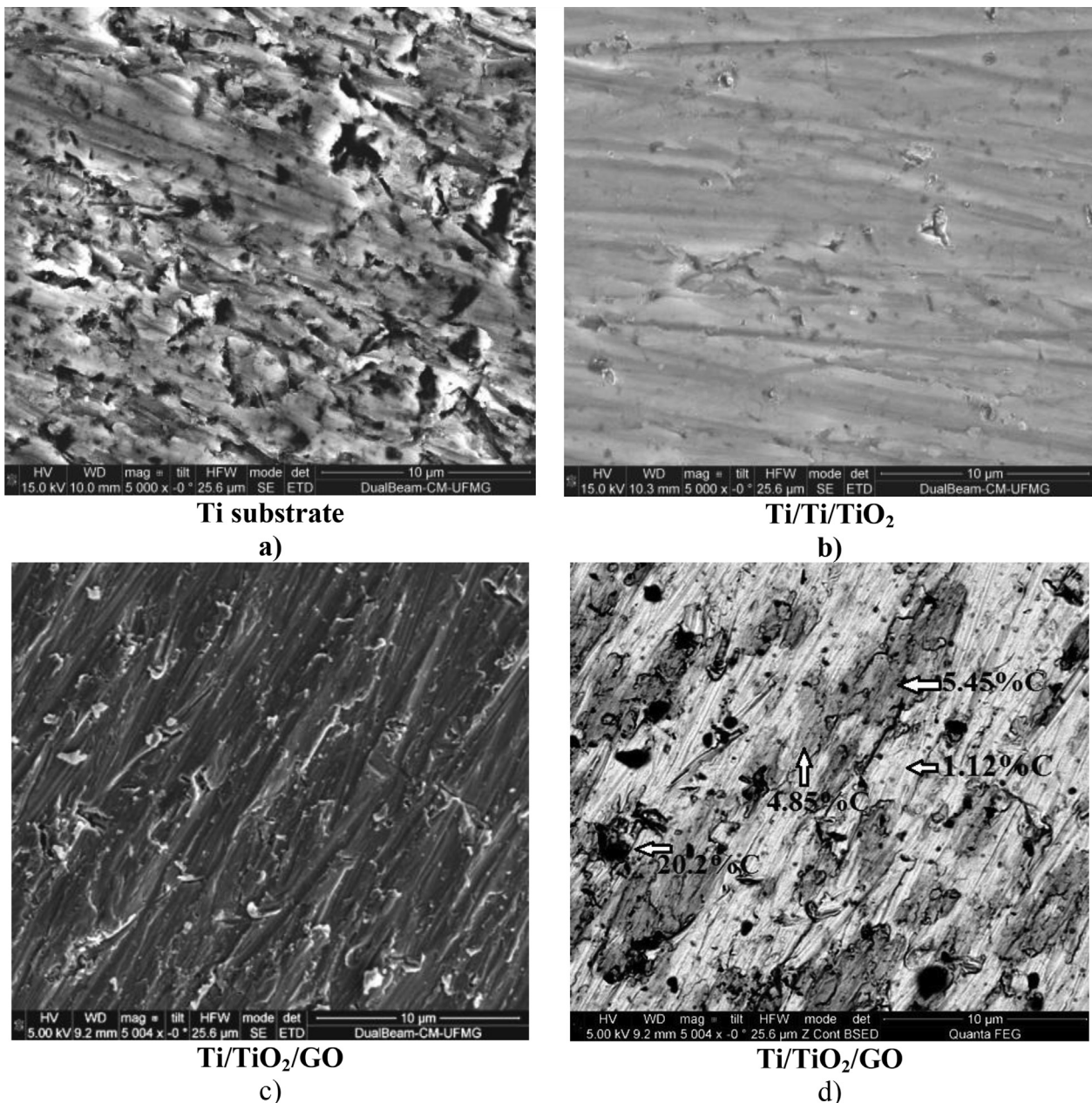


Fig. 4. SEM images of (a) Ti substrate, b) Ti/Ti/TiO₂, c) Ti/TiO₂/GO and d) backscattered electron image for Ti/TiO₂/GO with the carbon atomic percentage measured by EDS.

film composites is shown in the inset of Fig. 3. Thus, comparing this value with the I_D/I_G ratio values shown in Fig. 3, it can be deduced that the successive electrodeposition voltammetric cycles induced a slight decrease in the size of in-plane sp^2 domains and the expansion of the disorder in the prepared TiO_2/GO film.

The noise level discrepancy among the Raman spectra shown in Fig. 3(a) suggests that the film concentration on the substrate surface is not uniform. Thus, the composite film distribution on the Ti substrate was investigated by SEM; the results are shown in Fig. 4. The SEM image for the Ti-plate shows a surface characterized by many grooves and scratches, typical of abrasive wear. However, after film formation, the surface became smooth, as shown in Fig. 4(a–c) for Ti/Ti/TiO₂ and Ti/TiO₂/GO. Fig. 4(d) displays the backscattered electron image for Ti/TiO₂/GO. The color contrast observed in the backscattered electron image is generated by the different phase compositions relative to their average atomic number. Such differences in color distribution indicate a non-uniform film concentration on the surface, which is in agreement with the results obtained by Raman spectroscopy. The dark areas indicate the presence of lower atomic weight atoms such as C and O. Fig. 4 shows that the dark areas are located mainly in the Ti surface cracks. The GO distribution on this surface was investigated by measuring the carbon concentration by Energy-dispersive X-ray spectrometry (EDS). The measurements were collected from areas covering from light to dark regions of the surface. The EDS results are reported in Fig. 4(d). The highest carbon concentration values were found in the dark areas, which means that the film was deposited mainly in the cracks and scratches of Ti/TiO₂/GO.

The Raman spectrum region from 100 to 750 cm^{-1} was used to investigate the presence of TiO₂ [28,29]. The spectrum shown in Fig. 3(b) is very wide and exhibits a peak at $\sim 140\text{ cm}^{-1}$, observed for all samples and representative of anatase TiO₂, and a relatively broad background extending from 200 to 750 cm^{-1} . The broad continuous spectrum profile observed here is typical of amorphous semiconductors, demonstrating the predominantly amorphous structure of the oxide layer obtained by electrochemical deposition. However, sample Ti/Ti/TiO₂ showed two other bands with low intensity at 517 and 638 cm^{-1} , attributed to anatase TiO₂ [30]. These results are in agreement with those obtained by X-ray diffraction and confirm the presence of anatase TiO₂ in the prepared films.

The UV–Vis absorption properties of the as-prepared composite films on Ti substrates were measured by DRUV technique. The results are shown in Fig. 5(a). The films exhibited TiO₂ absorption characteristics with an absorption peak around 340 nm. After GO was introduced (Ti/TiO₂/GO), this peak shifted to 360 nm and the light absorbance range broadened to higher wavelength regions. Thus, the Ti/TiO₂/GO bandgap energy was reduced in relation to Ti/Ti/TiO₂ and a more efficient utilization of the solar spectrum was achieved. A similar trend has been described for other systems where GO was incorporated into TiO₂ [15–17].

Photoluminescence (PL) emission spectroscopy was used to study the transfer behavior of photo-induced electrons and holes in the composite films. In general, a decrease in the PL intensity results from a lower photo-induced electron–hole pair recombination rate and leads to photocatalytic activity increases. The measured PL-emission spectra of the films in the range of 400–850 nm are presented in Fig. 5(b). The spectra of all samples are centered around 468 nm and the intensity reduced from Ti/Ti/TiO₂ to Ti/TiO₂/GO, indicating an increase in the photocatalytic activity. The lower PL emission intensity of Ti/TiO₂/GO may be attributed to the GO action as an electron shuttle for TiO₂, preventing electron–hole recombination with a resulting higher photocatalytic activity [31].

Fig. 6(a,b) shows the photocatalytic activity results of the composite films in the degradation of Reactive Yellow 145. The

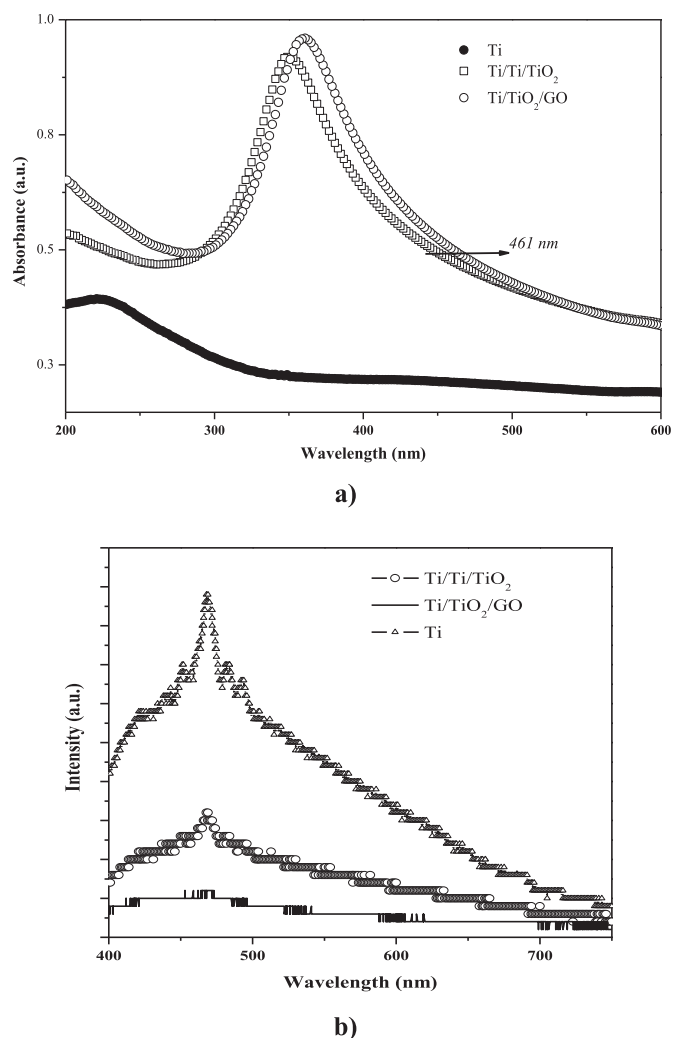


Fig. 5. a) UV–Vis absorption spectra and b) PL emission spectroscopy for the Ti substrate, Ti/Ti/TiO₂ and Ti/TiO₂/GO.

photocatalytic test was performed with a Reactive Yellow 145 synthetic solution and textile dyeing baths containing such a dye. In the case of the actual dyeing baths, other products are present besides the pigment, such as equalizing and fixing additives, acids or bases for pH control, salts and surfactants [32].

Fig. 6(a) shows that the dye concentration did not change in the first 10 min in the absence of light. The same result can be observed in Fig. 6(b) after 30 min in the dark. Similarly, no degradation occurred in the presence of the photocatalyst Ti/TiO₂/GO in the dark. On the other hand, both dye samples showed a small percentage of degradation when they were exposed to light, even in the absence of the photocatalyst.

The results depicted in Fig. 6(a,b) demonstrate that the TiO₂-based photocatalyst improved dye degradation in both samples. However, the as-prepared Ti/TiO₂/GO exhibited a higher photocatalytic activity, e.g., the rate of color removal in the first 5 min of the Reactive Yellow 145 synthetic solution in the presence of Ti/TiO₂/GO was 0.080 min^{-1} , while with Ti/Ti/TiO₂ it was only 0.045 min^{-1} . Similarly, for the textile dyeing baths, the rate of color removal employing Ti/TiO₂/GO as a catalyst was almost twice as high as with Ti/Ti/TiO₂. Thus, the addition of GO to the film improved light utilization in the degradation process by promoting

the separation of the electron–hole pairs and extending the electron life.

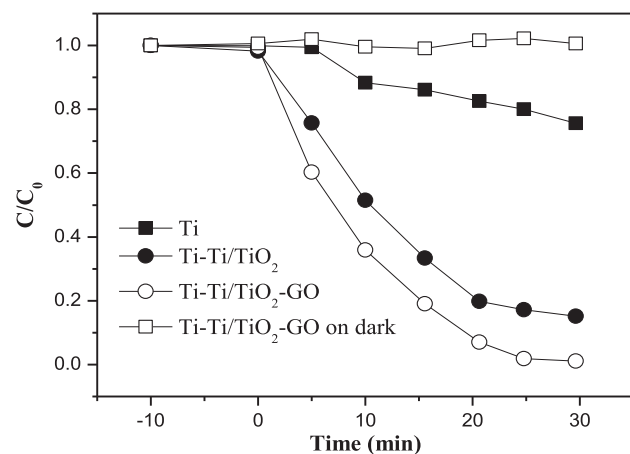
The stability and recyclability of photocatalysts Ti/TiO₂/GO and Ti/Ti/TiO₂ were evaluated over 15 cycles and the results are shown in Fig. 6(c–d). The exposure time was kept at 30 min. As already mentioned, the photocatalyst efficiency was lower in the absence of GO, but the behaviors of both films in the course of the different cycles were similar. Also, their photocatalytic activity reduced after the first cycle and remained on level in the subsequent cycles. However, Ti/TiO₂/GO was more efficient ($\approx 70\%$) even after the last cycles. This result demonstrates that the Ti/TiO₂/GO film is reasonably stable and can be employed several times.

The reduction in the photocatalytic activity observed for Ti/TiO₂/GO may have been due to GO leaching. The Raman spectra obtained at different points on the film surface region in the range from 100 to 750 cm⁻¹ were used to investigate the presence of GO in the support before and after various numbers of use cycles; the results are shown in Fig. 7. After the 12th cycle (Fig. 7(b)), the Raman spectra showed that the definition of the D and G bands was significantly reduced, suggesting that the GO concentration decreased after each application. However, it is interesting to note

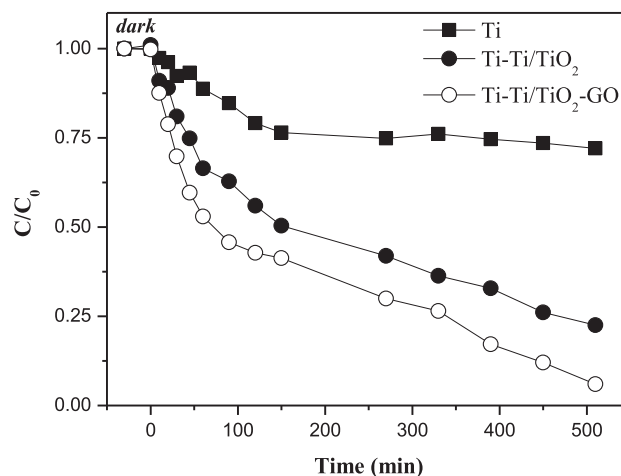
that GO was still present even after the 12th cycle, indicating a good adhesion between the film and the Ti substrate.

4. Conclusions

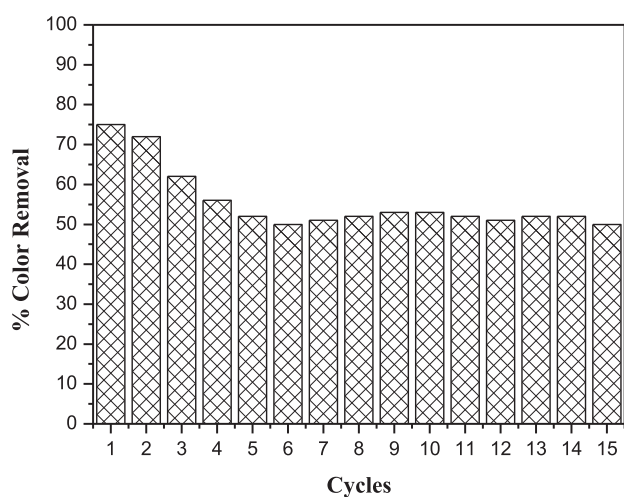
In summary, Ti/TiO₂/GO and Ti/Ti/TiO₂ composite films were prepared by electrodeposition onto a Ti substrate. Raman, SEM and EDS measurements showed that the film distribution on the substrate surface was not uniform, being deposited mainly into cracks and scratches. The as-prepared composite films exhibited characteristic TiO₂ absorption; however, Ti/TiO₂/GO exhibited a red shift on the band-edge at about 461 nm. The Ti/TiO₂/GO film showed the lowest PL emission intensity, which was associated with GO acting as an electron shuttle for TiO₂ with a resulting delay in the electron–hole recombination. The photocatalytic properties of the as-prepared films were investigated for the discoloration of Reactive Yellow 145 dye. The photocatalytic activity of Ti/TiO₂/GO was almost twice as high as that of Ti/Ti/TiO₂. In addition, Ti/TiO₂/GO presented a higher stability and recyclability, retaining its catalytic efficiency ($\approx 70\%$) even after 15 cycles.



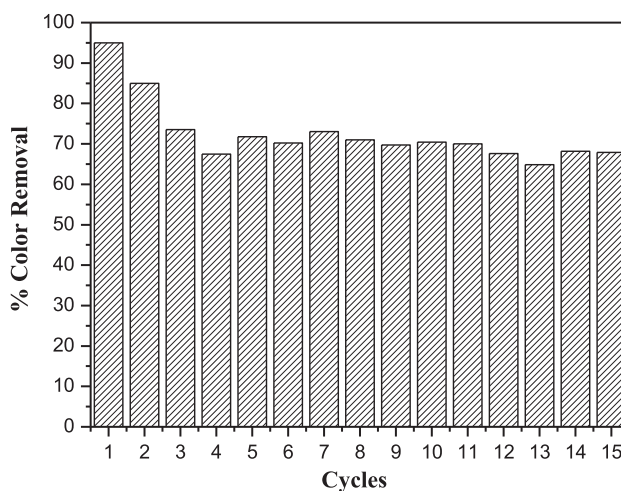
a)



b)



c)



d)

Fig. 6. Photocatalytic activity for the photodegradation of a) Reactive Yellow 145 synthetic solution, and b) the textile dyeing baths containing Reactive Yellow 145; Percentage of dye degradation in the Reactive Yellow 145 synthetic solution obtained over 15 film reuse cycles: c) Ti/Ti/TiO₂ and d) Ti/TiO₂/GO.

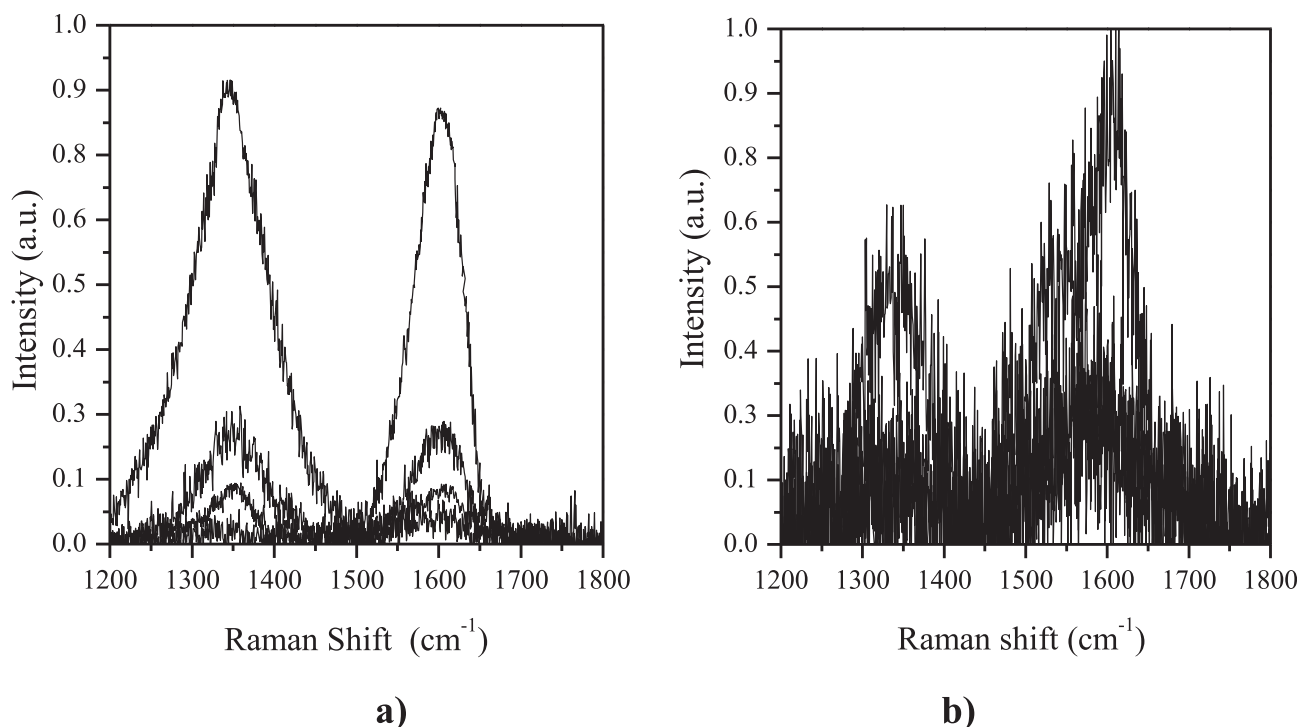


Fig. 7. Raman spectra of Ti/TiO₂/GO obtained at four different points on the film surface in the range of 100–750 cm⁻¹ (a) before and (b) after 15 use cycles.

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